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|    | <b>Experiment title:</b><br><i>In-situ</i> x-ray diffraction during sputter deposition of $Ti_{1-x}Al_xN$ – Part II | <b>Experiment number:</b><br><b>20_02_608</b> |
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### Report:

The **growth** of magnetron-sputtered  $Ti_{1-x}Al_xN$  thin films on amorphous substrates has been monitored by *in-situ* XRD. The **texture** development of these films shows a **strong dependence on deposition rate**, while **low values of the Al concentration x** and the deposition **temperature play a negligible role**. **Suppressing the ion current** to the substrate by a positive bias voltage can completely **reverse the final texture** even for low deposition rates, showing the importance of ion-substrate interaction, in accordance with [1].

### EXPERIMENTAL

The  $Ti_{1-x}Al_xN$  films have been grown in the ROBL deposition chamber, the characteristics of which are described in detail elsewhere [2]. As substrates, Si(100) wafers with a 1400 Å amorphous oxide capping layer of size 15 x 15 mm have been used. The base pressure before deposition was appr.  $1 \times 10^{-4}$  Pa. Ar and  $N_2$  were applied as sputter gases with a gas flow of 2.82 sccm, leading to a constant working pressure of 0.35 Pa. During the reactive co-sputtering process the Ti and Al targets of 99.999 % purity were run at a constant DC power of 80 W, and 0/5 W, respectively. For a reactive gas mixture of  $Ar/N_2 = 2/0.6$  this resulted in a growth rate of  $\sim 1$  Å/s, and halved down to 0.5 Å/s for a mixture of  $Ar/N_2 = 2/1$ . The substrate temperature was varied between 150°C and 300°C. A bias voltage of -30 V and +10 V was applied.

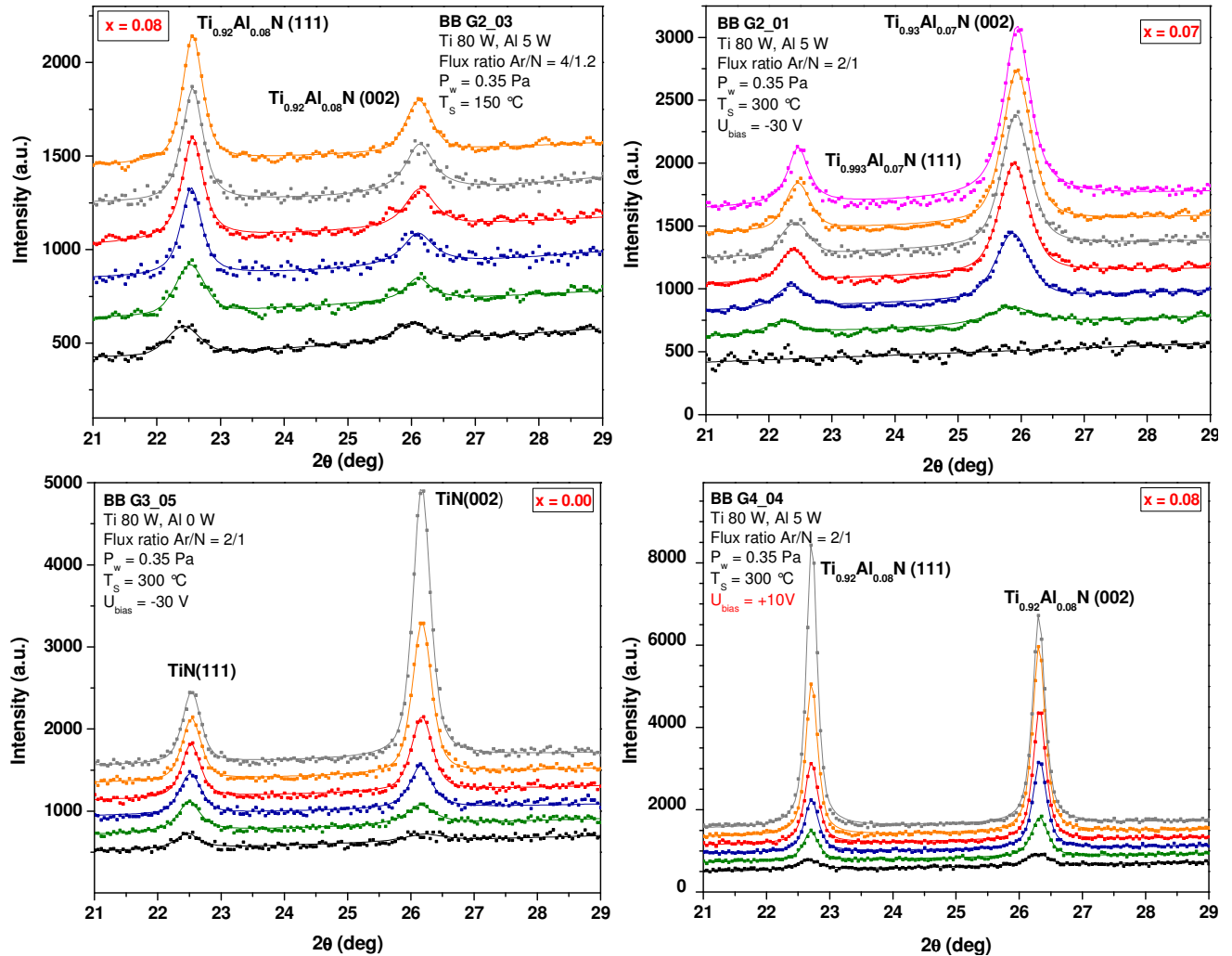
The growth process has been studied *in-situ* by three different scattering techniques. First, specular relectivity (XRR) for growth rate calculation. Second and third , large angle off-plane scattering in Bragg-Brentano geometry (XRD) and Grazing Incidence Grazing exit large angle in-plane Scattering (GIXS). The integrated peak intensities reveal information about the texture and crystallinity, and the exact positions of the Bragg peaks yield information about the lattice constants, off-plane and in-plane, repectively.

For all samples, the energy of the incident x-rays was monochromatized to 12.917 keV ( $\lambda = 0.961$  Å).

|       | Power Ti (W) | Power Al (W) | Temperature (°C) | Gas ratio Ar/N | Bias (V) |
|-------|--------------|--------------|------------------|----------------|----------|
| G3_05 | 80           | 0            | 300              | 2/1            | - 30     |
| G2_01 | 80           | 5            | 300              | 2/1            | - 30     |
| G2_02 | 80           | 5            | 300              | 2/1            | - 30     |
| G2_03 | 80           | 5            | 150              | 4/1.2          | - 30     |
| G4_04 | 80           | 5            | 300              | 2/1            | + 10     |

Deposition parameters for the various samples investigated.

## RESULTS



**Fig 1:** The dotted curves show the observed Bragg-Brentano intensities with increasing sample thickness (final thickness around 200–300 nm), the superposed straight lines show the fits obtained using a Pseudo-Voigt function for each single peak. The GIXS data show corresponding behaviour (*not shown*).

A deposition at 150°C and a growth rate of 1 Å/s results in a typical (111) preferred off-plane orientation for higher thicknesses (Fig. 1 *top left*). An increase of nitrogen partial pressure leads to stronger target poisoning which in consequence leads to a lower deposition rate of 0.5 Å/s. This strongly effects the preferred off-plane orientation, which stays (002) even at higher thicknesses, practically regardless of substrate temperature (Fig. 1 *top right*). This behaviour cannot be attributed to the small amount of Al, since pure TiN shows exactly the same behaviour (Fig. 1 *bottom left*).

Because a reduced deposition rate also changes the ion/neutral-flux-ratio on the substrate, we suggest that the change in preferred orientation is not due to strain surface-energy competition but due to the increased ion bombardement during growth [3], and in fact suppressing the ion current by applying a positive bias reverses the texture into a competitive growth, where (111) orientation gets the upper hand at higher thicknesses (Fig. 1 *bottom right*).

- [1] D. Gall, S. Kodambaka, M.A. Wall, I. Petrov, and J.E. Greene, *J. Appl. Phys.* **93** (2003) 9086
- [2] W. Matz, N. Schell, W. Neumann, J. Böttiger, and J. Chevallier, *Rev. Sci. Instrum.* **72** (2001) 3344
- [3] M. Beckers, N. Schell, R.M.S. Martins, A. Mücklich, and W. Möller, *submitted* for publication