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Report:

In a continuation of experiment Si-1499, we have completed a systematical study of InAscore/InAsP-shell nanowire structures. The samples were grown by chemical vapor deposition at Lund University (Sweden) on Si(111) substrates [1,2], using an allyl alcohol layer to obtain wire nucleation [2]. Using arsine and trimethylindium as precursor gases, InAs wires are grown at a substrate temperatures around 550°C. These InAs cores have a nominal thickness of 100 nm and a length of about $2 \mu m$. Subsequently, phosphine is added as precursor to cover the wires by a shell of InAsP: two sample series have been fabricated, one with pure InP shells wof different thicknesses, and a second one where the arsine/phophine ratio was varied to obtain shells with different InP content, and hence different mismatch between core and shell.

For all samples, high resolution reciprocal space maps have been recorded both in coplanar diffraction geometry as well as in grazing incidence diffraction (GID) geometry. Hence the lattice parameters parallel and perpendicular to the growth plane have been determined to obtain the material composition from the unit cell strains. The main question addressed here is the strain state of the shell, and in particular whether the shell is grown pseudomorphically, or plastic relaxation plays already a role: the latter is expected to lead to a deterioration of optoelectronic properties due to incorporation of extended defects.

In contrast to a pseudomorphic 2D layer, the strain state of a pseudomorphic shell wrapped around a core with hexagonal shape is rather complicated. Furthermore, since the shell thickness and the core diameter are of the same order of magnitude at least for the thicker shells in the investigated samples, straining of the core has to be taken into account. For this reason simulations of the RSMs have been performed, starting from models of the wires and calculating their strain state using a finite element method (FEM) package. Kinematical scattering theory was used to simulate the corresponding RSMs: due to the small size and high aspect ratio, multiple scattering effects do not need to be taken into account in the simulations even in GID geometry.

Figure 1 shows as colormaps RSMs for several samples of the two series. The best simulations are overlaid as black contour lines. Obviously, while for low InP contents and thin shells a good match is obtained, for the thicker shells and/or the higher InP contents, no good correspondence is found between experiment and simulation. This indicates that in this case plastic relaxation is already important. From these data, the critical shell thickness and/or shell mismatch can be determined, and the growth process optimized accordingly: it is found that the growth rates of the shell differ significantly as compared to growth rates calibrated using 2D layer gorwth. Detailed data analysis currently under way shall also reveal whether alloying between core and shell is important in these samples.



Fig. 1: Reciprocal space maps of two InAs-core/InAsP-shell wire sample series (colormaps) and best fitting simulations (black contour lines). Excellent correspondence to model calculation is found for thin shells and low InP contents, while otherwise no fitting simulation could be achieved, indicating that plastic relaxation sets in at the corresponding strain and thickness values.

References

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- [2] Th. Mårtensson, J.B. Wagner, E. Hilner, A. Mikkelsen, C. Thelander, J. Stangl, B.J. Ohlsson, A. Gustafsson, E. Lundgren, L. Samuelson, W. Seifert, Advanced Materials 19, 1801 (2007).