

3D grain structures from X-ray diffraction contrast tomography

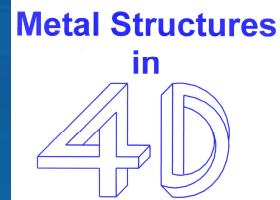
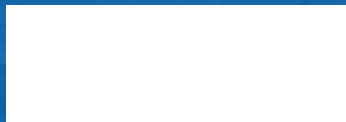
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MRS BULLETIN

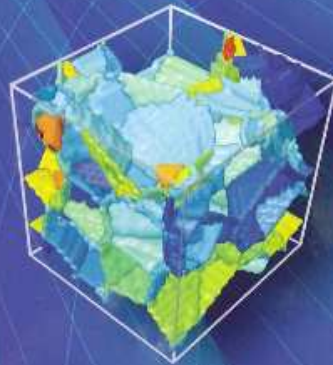
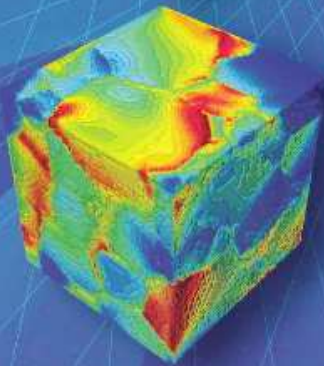
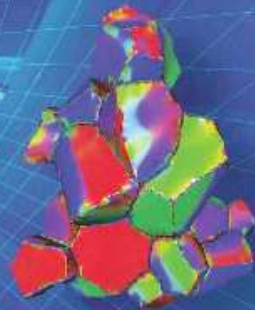
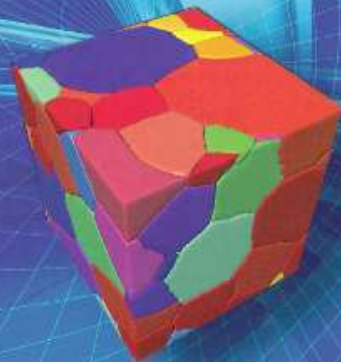
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Three-Dimensional Materials Science



Improve our understanding
of how structural materials
react to external stimuli

- deformation
- annealing
- chemical environment

Focus on metallic alloys
(polycrystalline)

Spanos, Rowenhorst, Lewis, Geltmacher
*Combining serial sectioning, EBSD and
Finite Element Modeling, MRS Bulletin*



Topotomography of Al grain
during recrystallization

Outline:

Non destructive mapping of polycrystals in 3D

Diffraction contrast tomography:

- principle
- analysis strategy
- current possibilities
- limitations

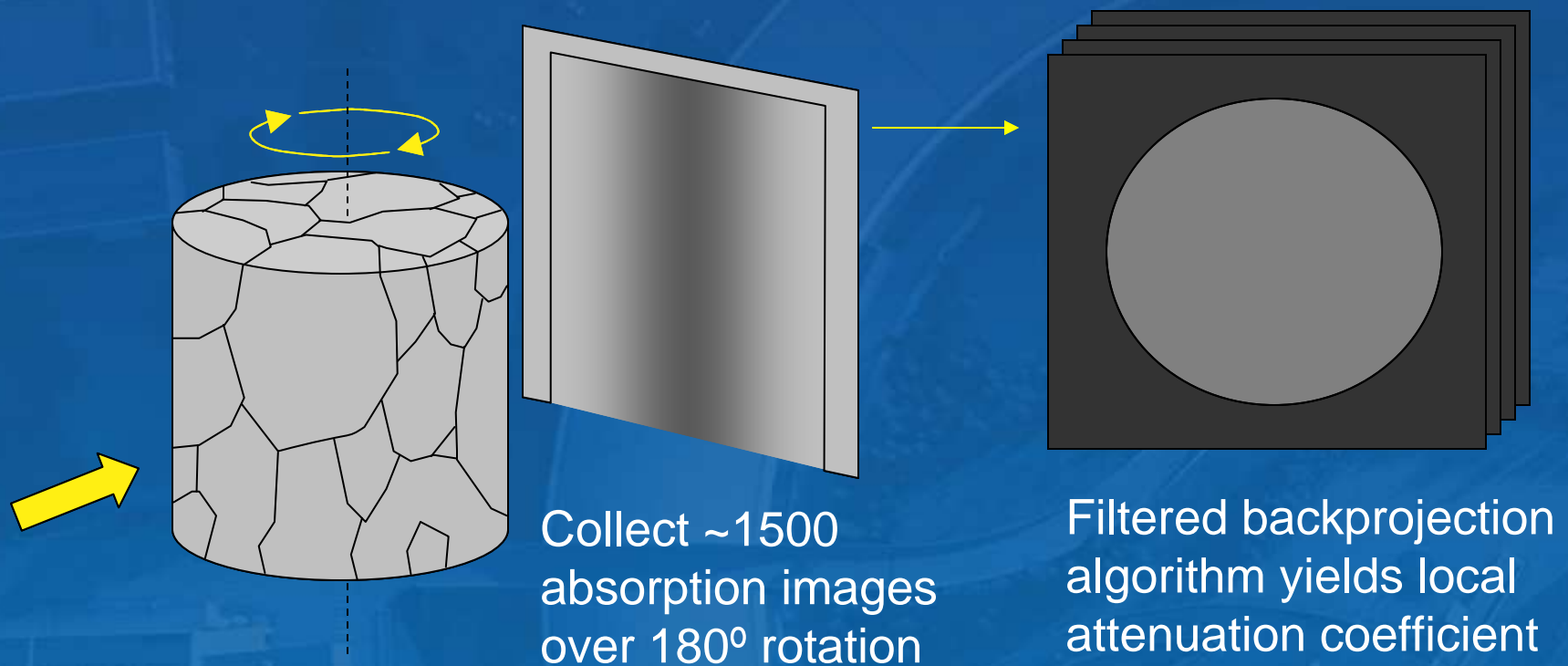
Applications

Conclusions

Conventional absorption tomography

Synchrotron or lab X-rays, neutrons, etc

- No contrast between grains of the same phase
- No crystallographic orientation or strain information



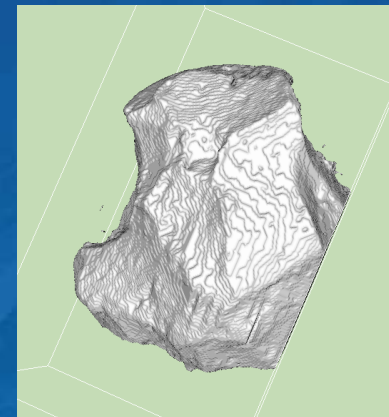
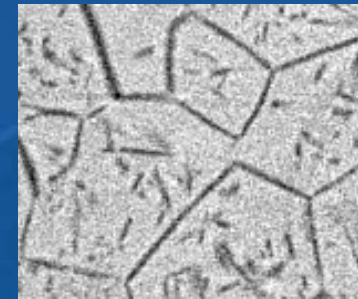
...some exceptions

- Segregation and/or precipitation at grain boundaries
- Duplex microstructures
- Liquid metal penetration (Al - Ga)

X-ray absorption or phase contrast imaging may reveal 3D grain structure in these cases

However: crystallographic orientation unknown

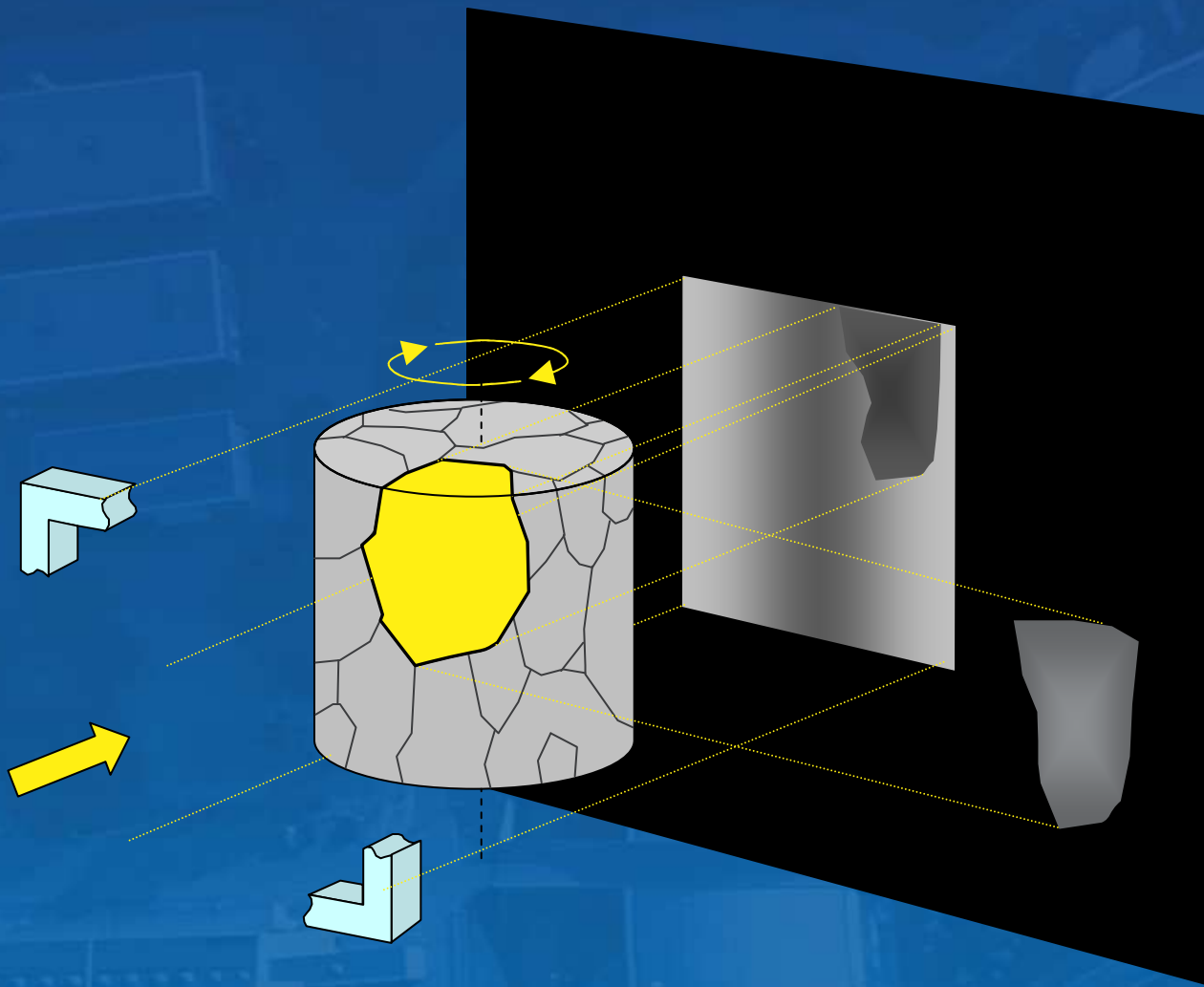
3D grain microstructure from X-ray phase contrast tomography



Ti β alloy (21s) with α phase precipitates

E.M. Lauridsen, R. Fonda, W. Ludwig et al.

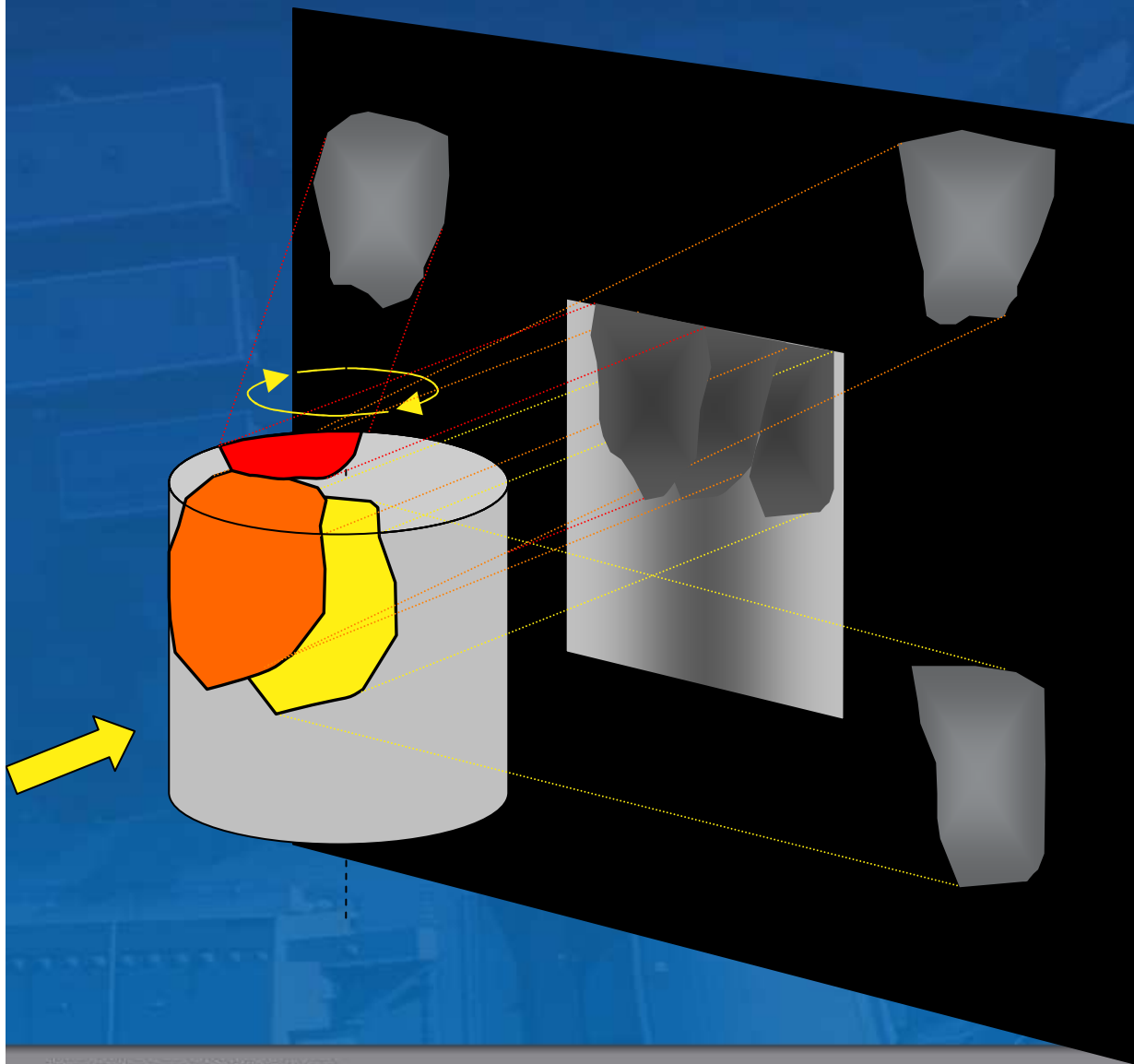
Diffraction Contrast Tomography: combined case



- Conventional tomo-
- During rotation, grains pass through diffracting alignments
- Large detector with high dynamic range
- grains with small spreads in orientation:
- “extinction” spot visible in direct beam
- Use slits to confine the beam to sample
- both spots can be approximated as grain projections ($\Delta\lambda/\lambda \sim 10^{-4}$)

G. Johnson, A. King, M. Hoennicke, T. Marrow, W. Ludwig, J. Appl. Cryst. 2008

Diffraction contrast tomography: combined acquisition



During sample rotation should see each grain ~ 20 - 100 times.

Some may be lost

- Overlaps
- Off the detector
- Low contrast

Enough to reveal the 3D grain shape through ART reconstruction

Determination of diffraction vectors

Segment and record diffraction spots in database ($\sim 20,000 - 100,000$)

Need to determine diffraction angles (θ, η, ω)

Find Friedel pairs of diffraction spots

- $(hkl), (-h-k-l)$ separated by $180^\circ \omega$

Visualise 180° sample rotation as reversed beam direction, and detector position.

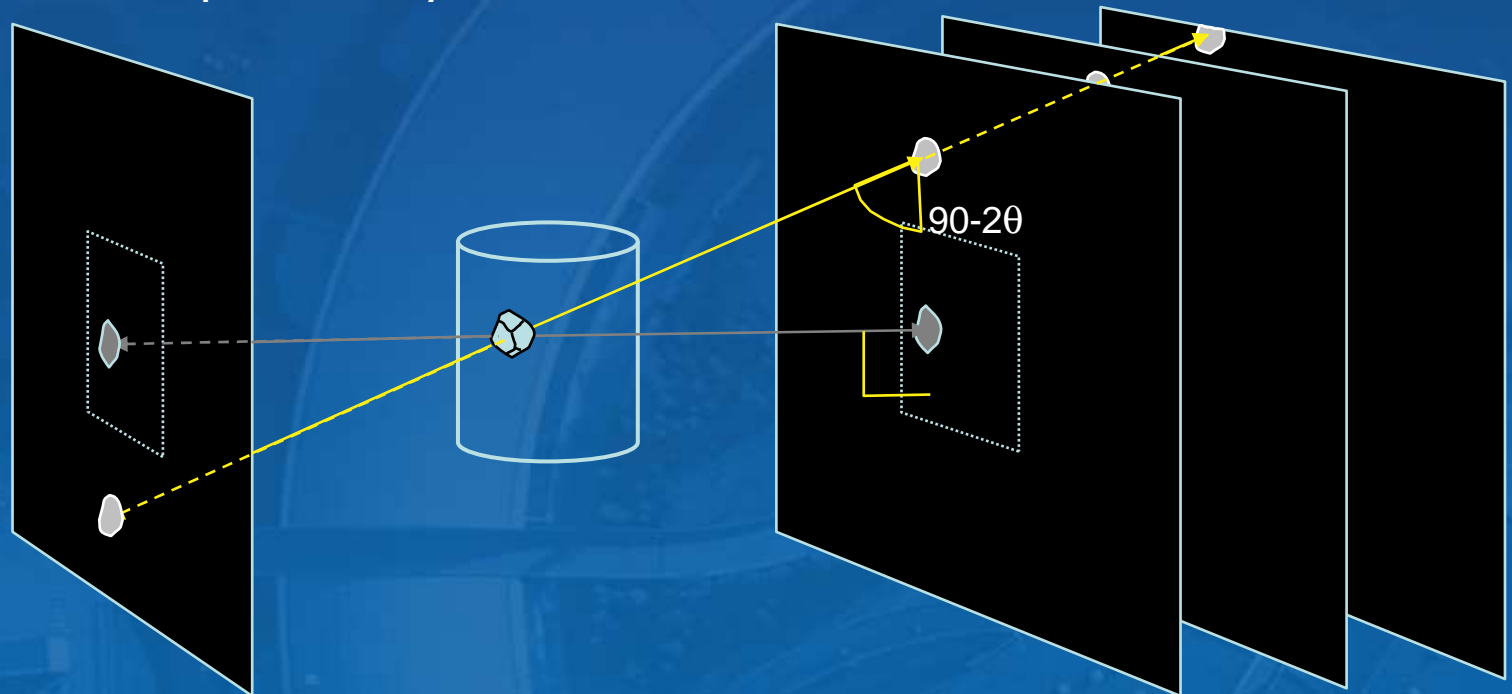
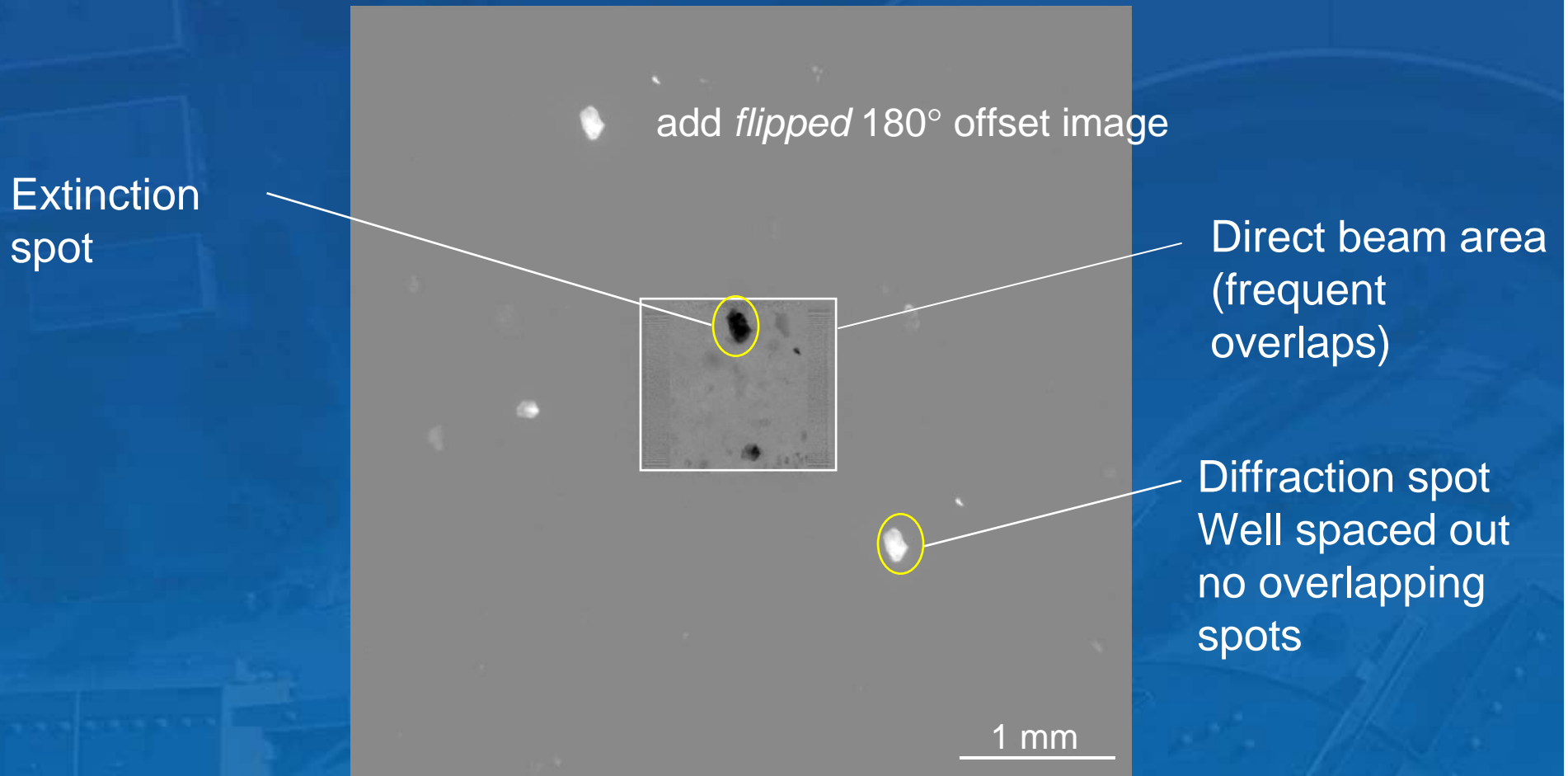


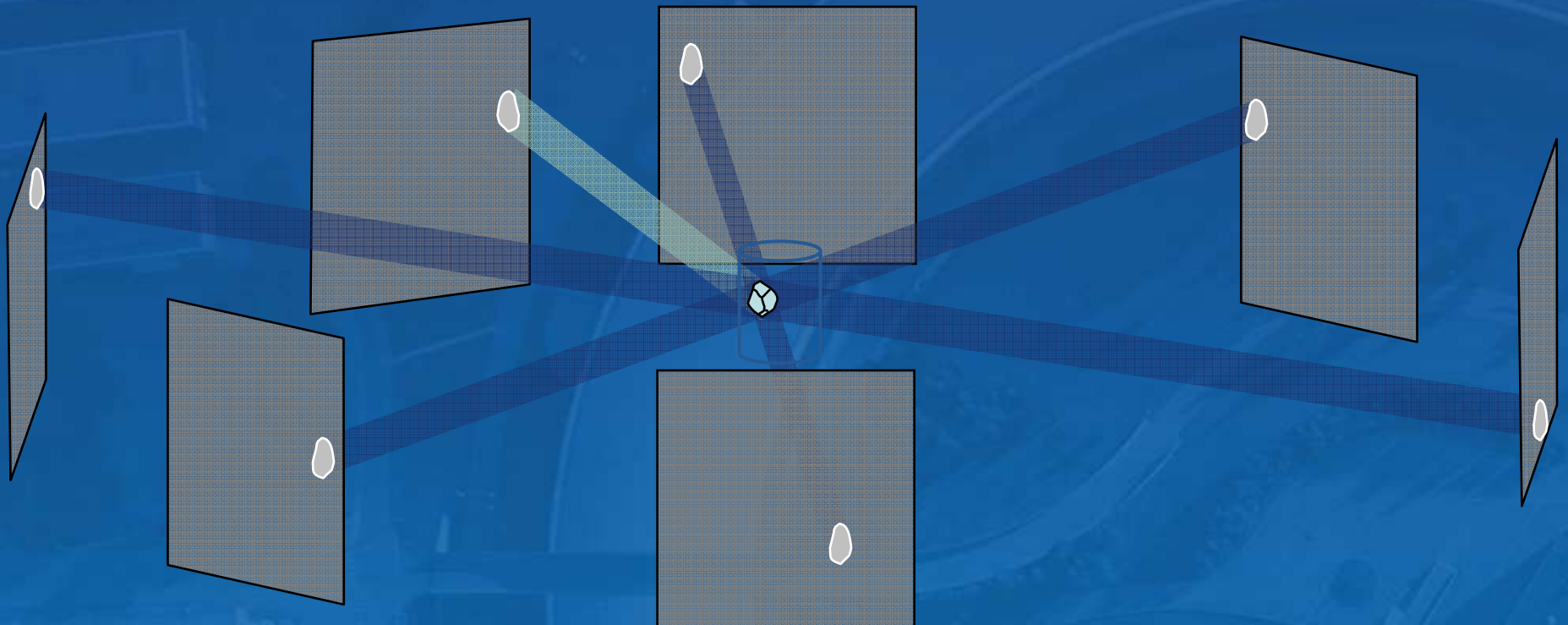
Image processing

Separate diffraction contrasts from images
(background removal, filtering)



3D backprojection geometry (as opposed to pseudo 2D) for the ART reconstruction has been implemented

- Gives more accurate grain reconstructions



Introduction - Principles - Results - Perspectives - Conclusions

Analysis route

1. Background removal, integration and segmentation of diffraction spots
2. Find Friedel pairs of diffraction spots (pair matching)
3. Find consistent groups of reflections (indexing)

Results:

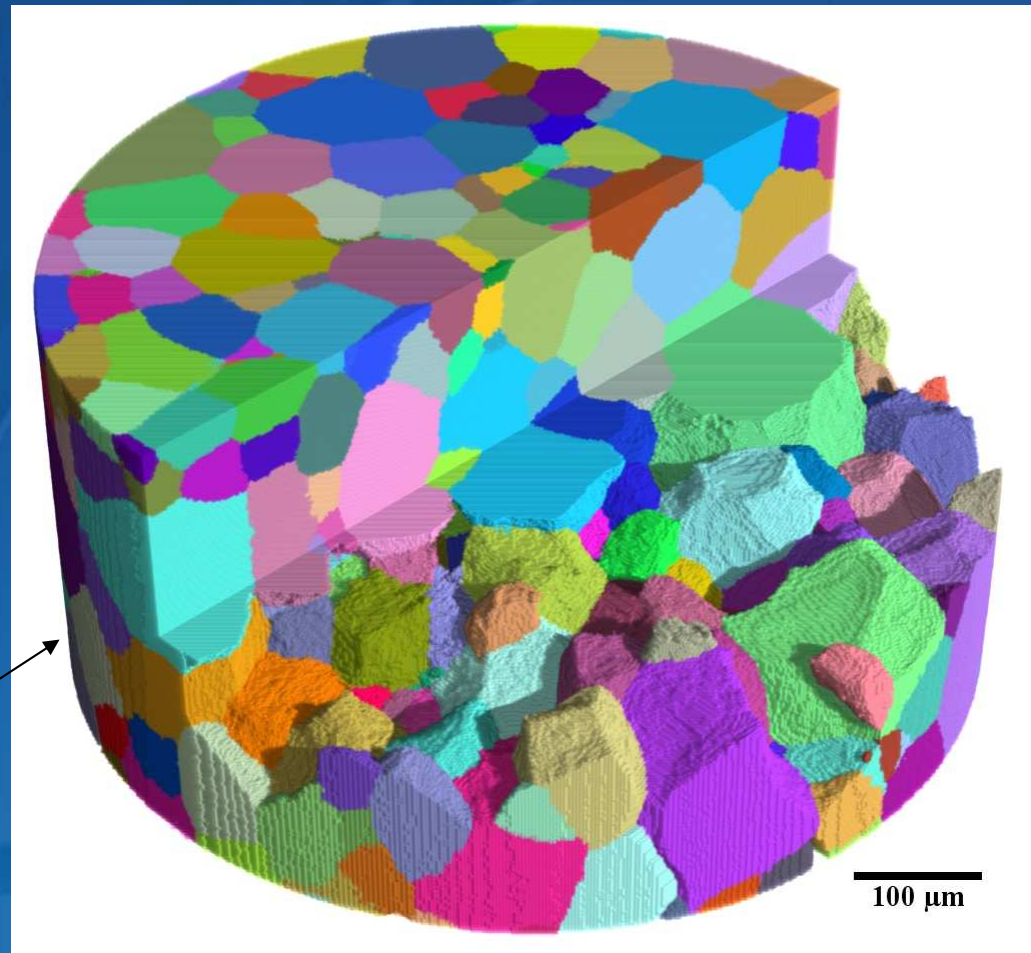
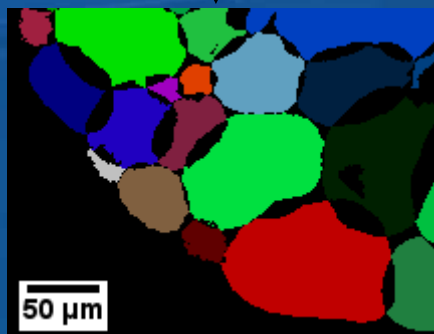
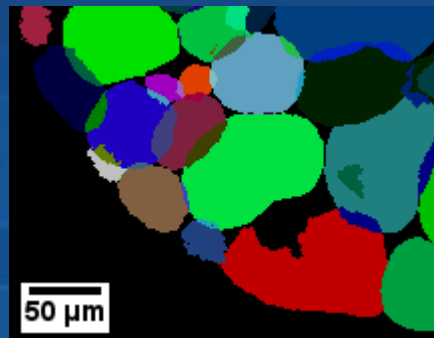
- Sets of projections belonging to individual grains
- Grain orientation
- Grain position
- elastic strain tensors (optional)

QuickTime™ and a decompressor are needed to see this picture.

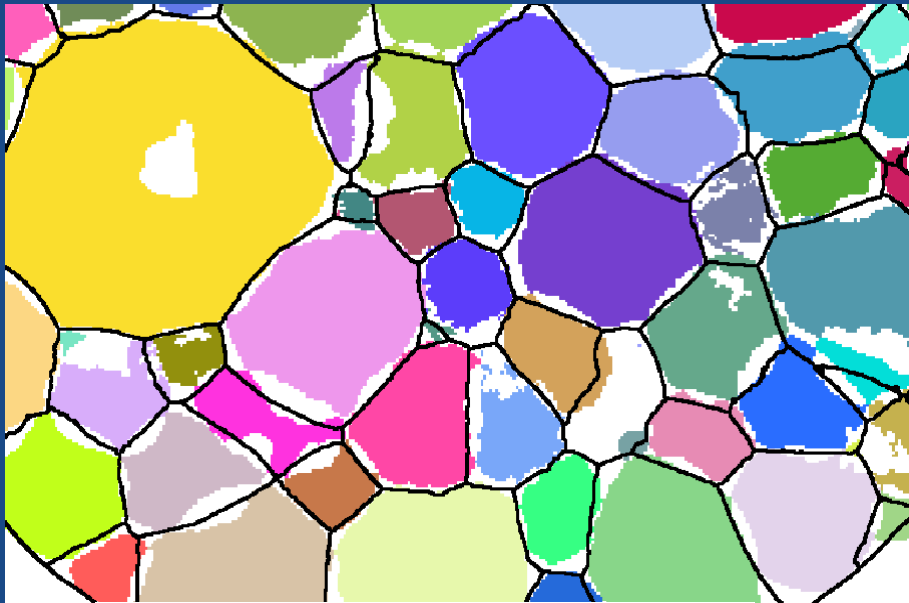
5. 3D grain shape from ART (grain by grain)
6. Assemble sample volume
7. Post-process grain map (remove overlaps, fill gaps)

200 μm

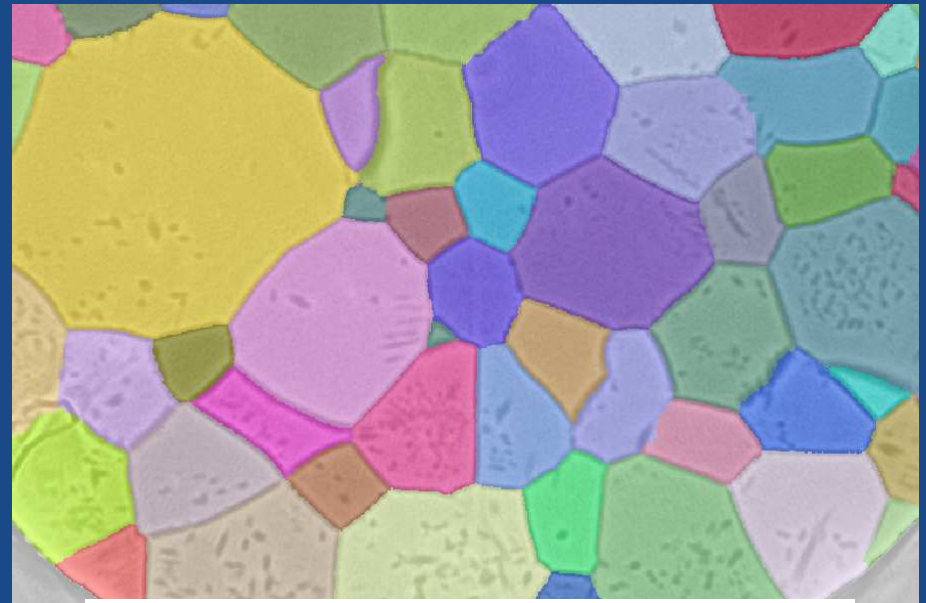
Experimental – Diffraction Contrast Tomography



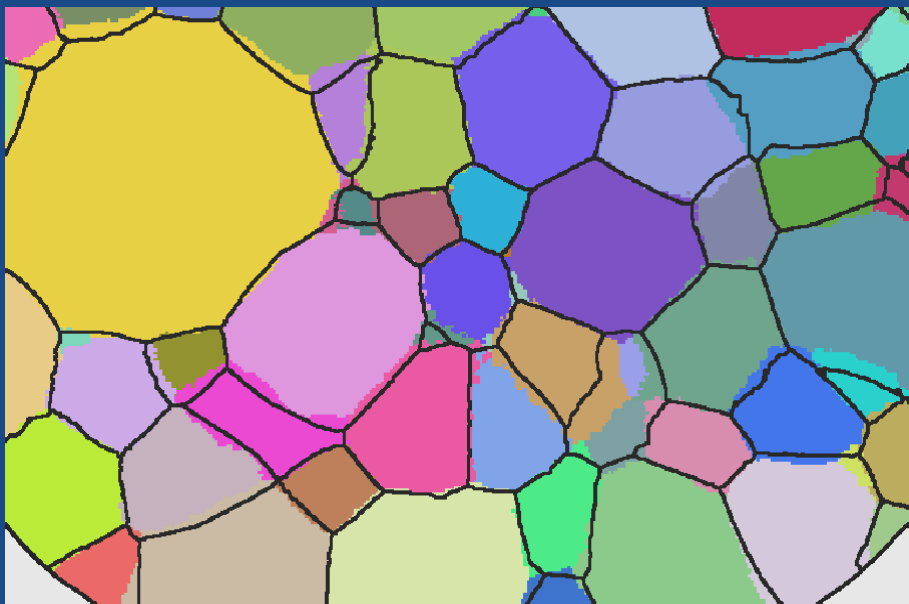
Question: are grain boundaries correct?



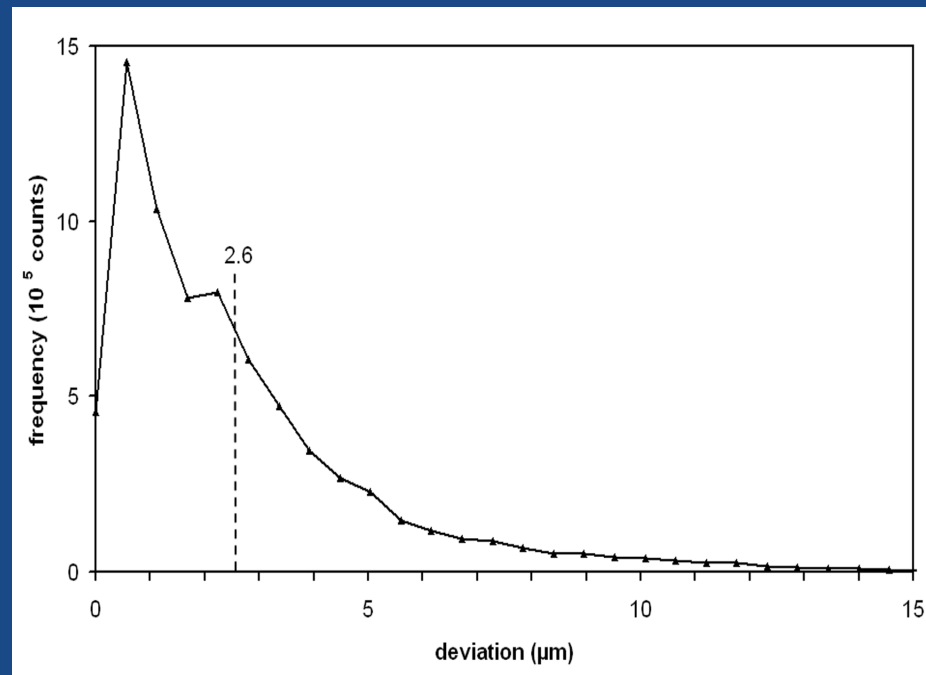
DCT raw reconstruction



Phase contrast tomography



DCT after postprocessing



3D grain reconstruction from DCT

QuickTime™ and a decompressor are needed to see this picture.

QuickTime™ and a decompressor are needed to see this picture.

100 μm

- Acquisition time @ ID11: 2 h
- 1008 reconstructed grains
- 2 days of processing (30 nodes)
- ~ 3 μm accuracy

W. Ludwig et al., Rev. Sci. Instrum. **80** 033905 (2009)

Known limitations...

1. Sample requirements

- low mosaicity (< 1 degree: recrystallisation or solidification)
- grain size bigger than about 20 times pixel size
- sample diameter less than 20 times grain size
- strong texture and high mosaicity reduce these numbers

2. Significant orientation & strain distributions inside grains violate assumption of parallel projections

3. Assumption of kinematical scattering may be violated: Absorption, multiple scattering & dynamical diffraction effects alter intensity distributions

Application examples

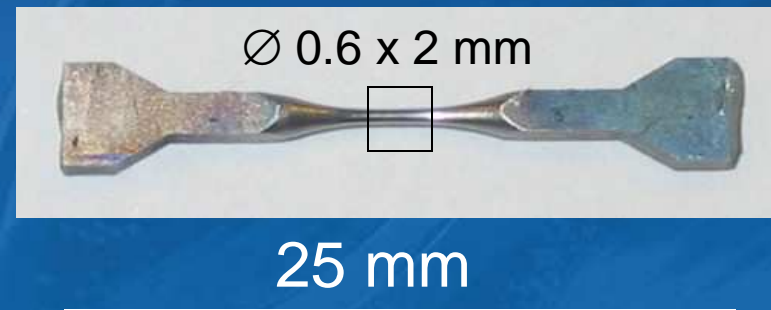
Determination of elastic strain tensors
during a tensile load test
(master thesis P. Reischig, TU Delft 2008)

Analysis of short fatigue crack
propagation in Ti alloy
(PhD thesis project of M. Herbig)

Measurement device



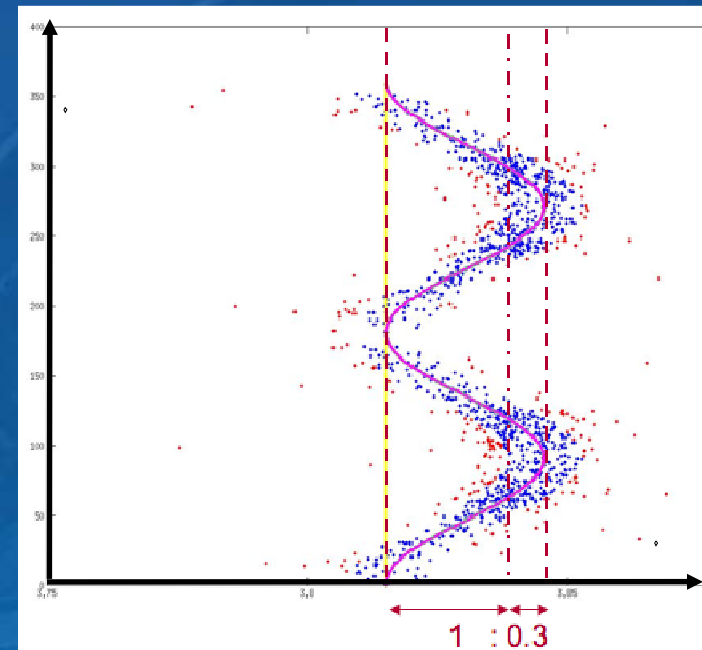
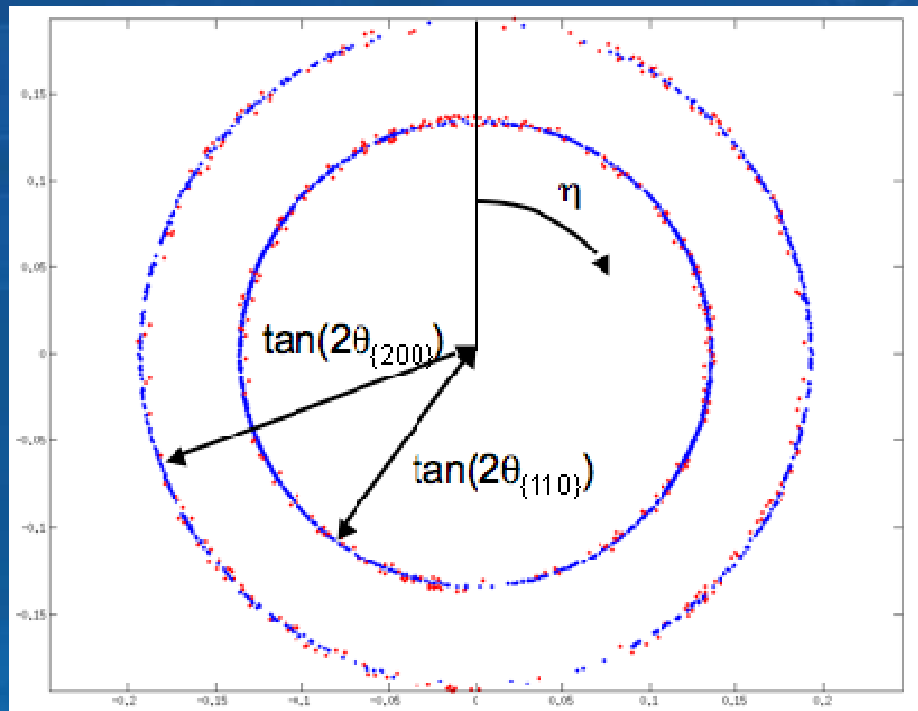
- 360 degree visibility
- 500N load cell
- mechanical loading (fine thread)



Material: Ti β alloy 21S (bcc)
 electrical discharge machining

“Artificial” powder diffraction pattern

extracted diffraction angles from the Friedel pairs
no dependence on grain position



clear angular shift
due to applied strain

Fit of strain tensors using linear elasticity model

Combines:

change in interplanar angles
change in interplanar distance

$$\begin{bmatrix} \Delta(\mathbf{n}_i \cdot \mathbf{n}_j) \\ \epsilon_{n_i}^{rev} \end{bmatrix} = \begin{bmatrix} M_{(n_i, n_j)} \\ L_{(n_i, n_i)} \end{bmatrix} \cdot \langle \epsilon^{rev} \rangle$$

Input

Measured variables: \mathbf{n} (plane normals), Bragg angles
Reference lattice parameter

Theoretical interplanar angles in unstrained state (bcc)

P. Reischig, master thesis

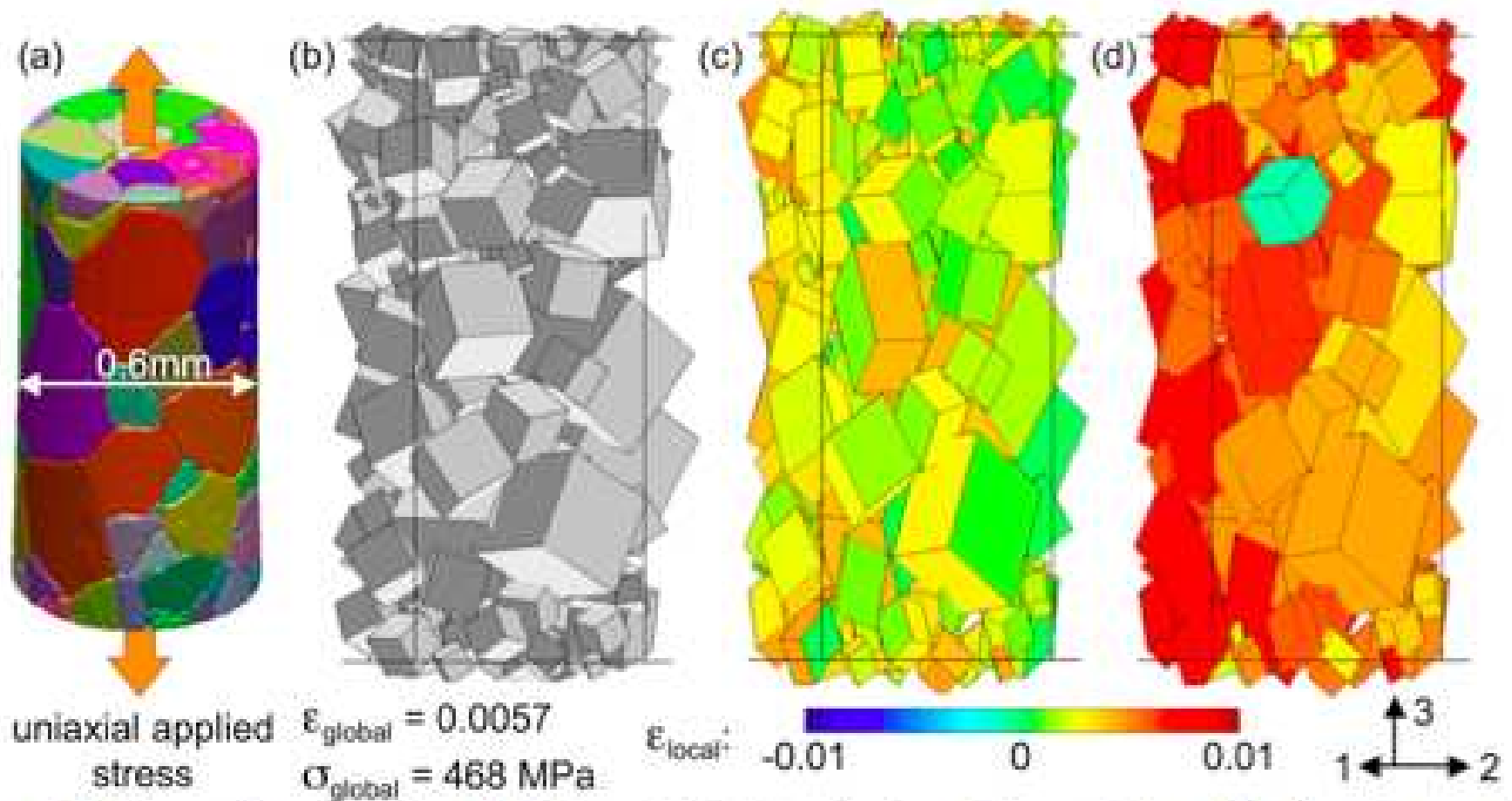
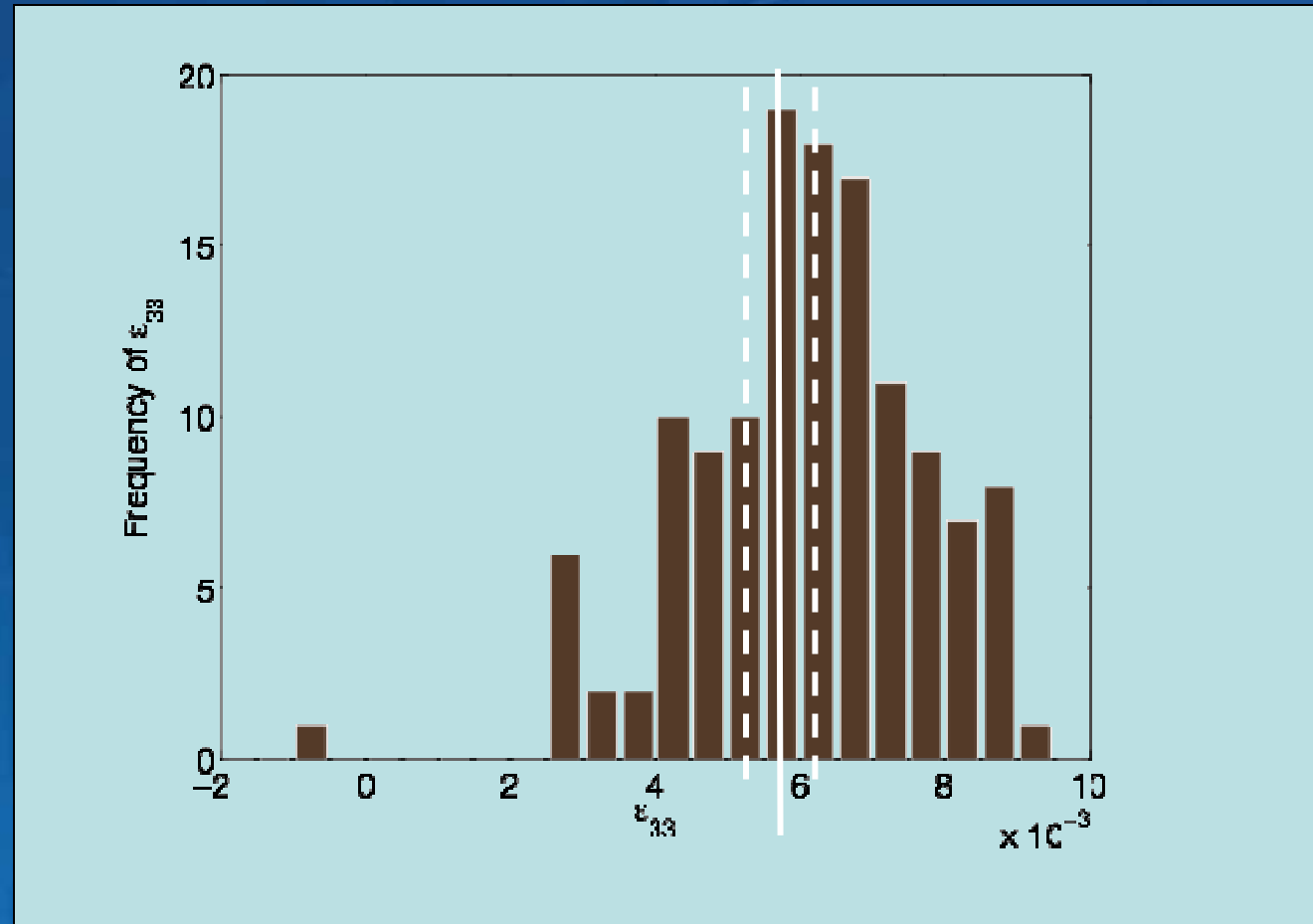


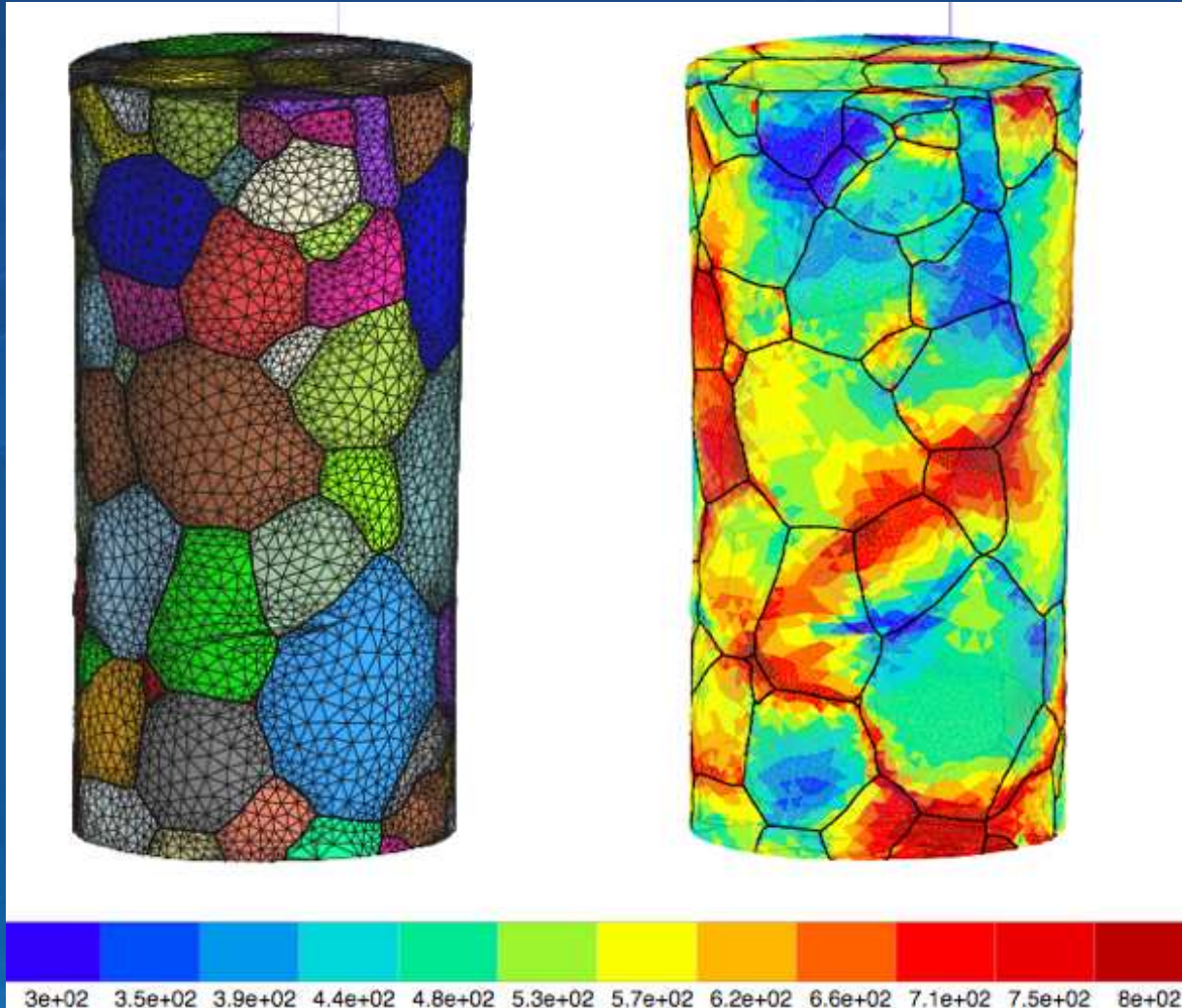
Figure B. Information currently available from DCT: (a) grain shapes (b) crystallographic orientations, and strain tensors, shown (c) resolved onto $\{002\}$ lattice planes and (d) strain parallel to tensile axis.

Elongation along the load axis (ϵ_{33})

$\sigma = 467.9 \text{ Mpa}$
 $\epsilon = 0.00572$



... analysis in progress



Strong variation
of local stress
& strain distribution

Comparison of grain
average values:
...in progress

Need to develop
local characterization
techniques
(e.g. combine DCT &
scanning μ -diffraction)

Collaboration with H. Proudhon, S. Forest, ENSMP

Characterization of short fatigue crack propagation

“Short fatigue crack problem”

- strong variability
- no reliable prediction / simulation
- need for more fundamental work
- experimental challenge

QuickTime™ and a decompressor are needed to see this picture.

SEM: Fatigue fracture surface

- Experiment:
- DCT of FIB notched sample (ID11)
 - in-situ crack propagation (25 time steps, ID19)

Heat treatment (grain boundary decoration)

- holotomography (ID19)

In situ crack propagation at ID19

Crack initiation from
FIB notch

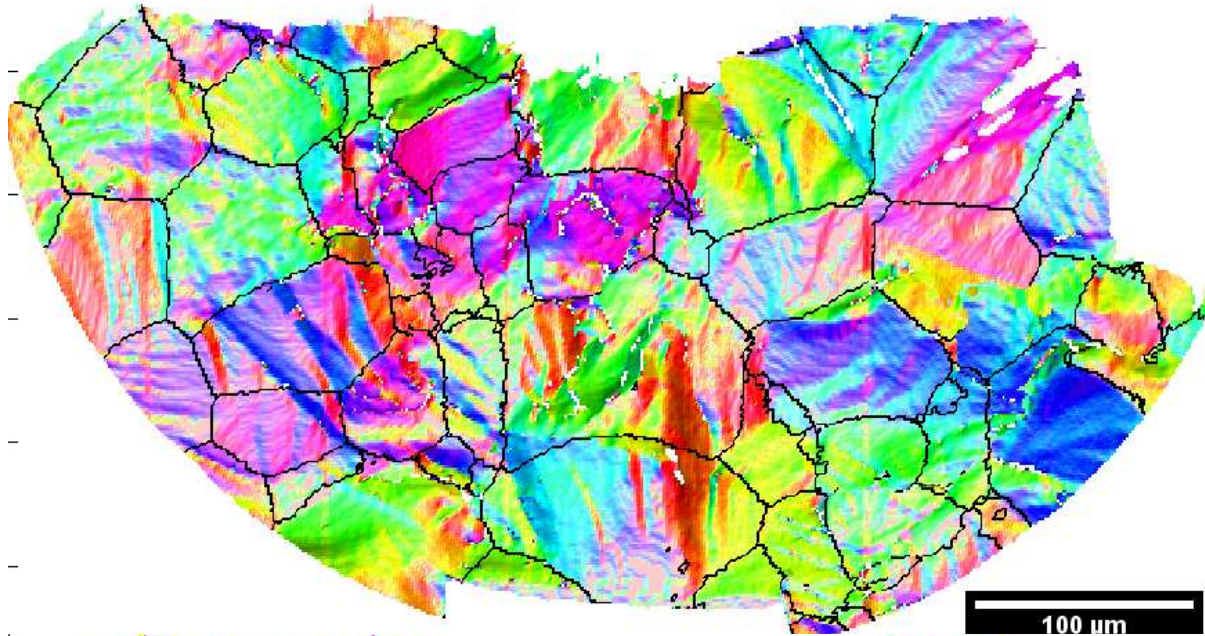
25 time steps

0.7 μm voxels

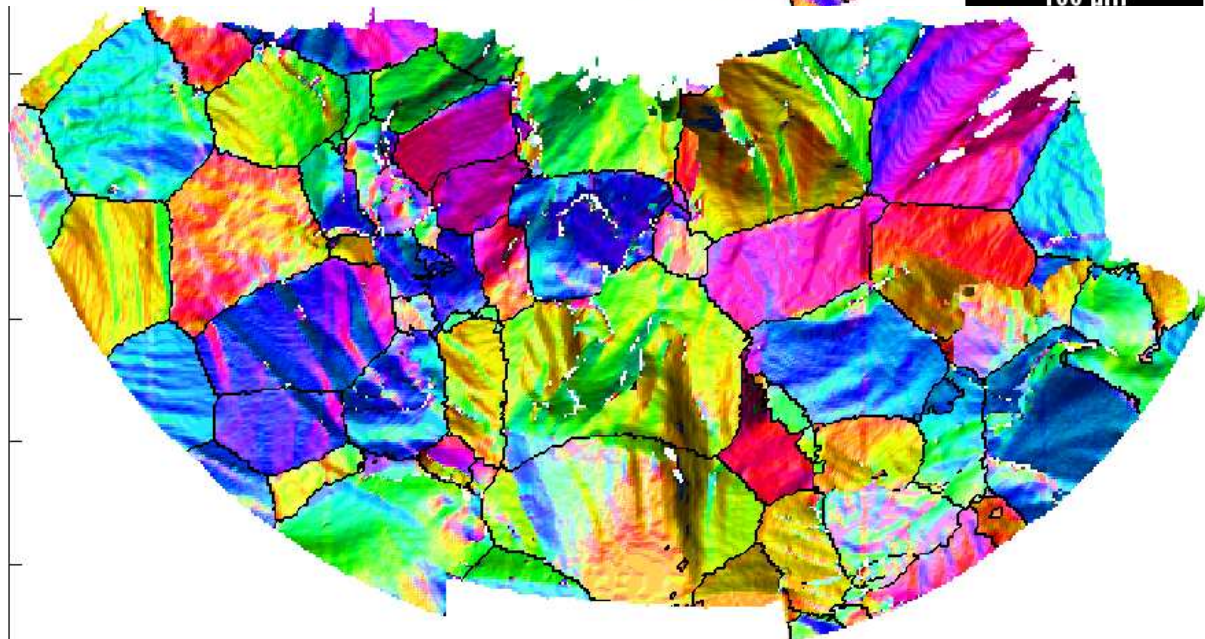
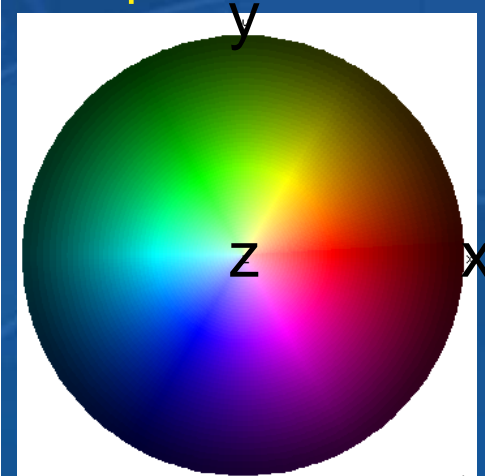
QuickTime™ and a
decompressor
are needed to see this picture.

3D rendition of short fatigue crack in Ti 21S alloy sample

QuickTime™ and a
decompressor
are needed to see this picture.



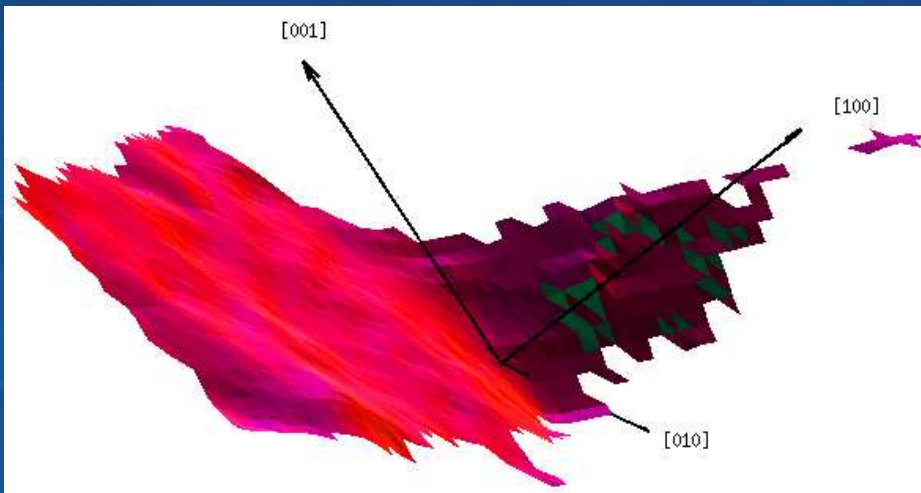
Orientation of crack in **sample coordinate** system



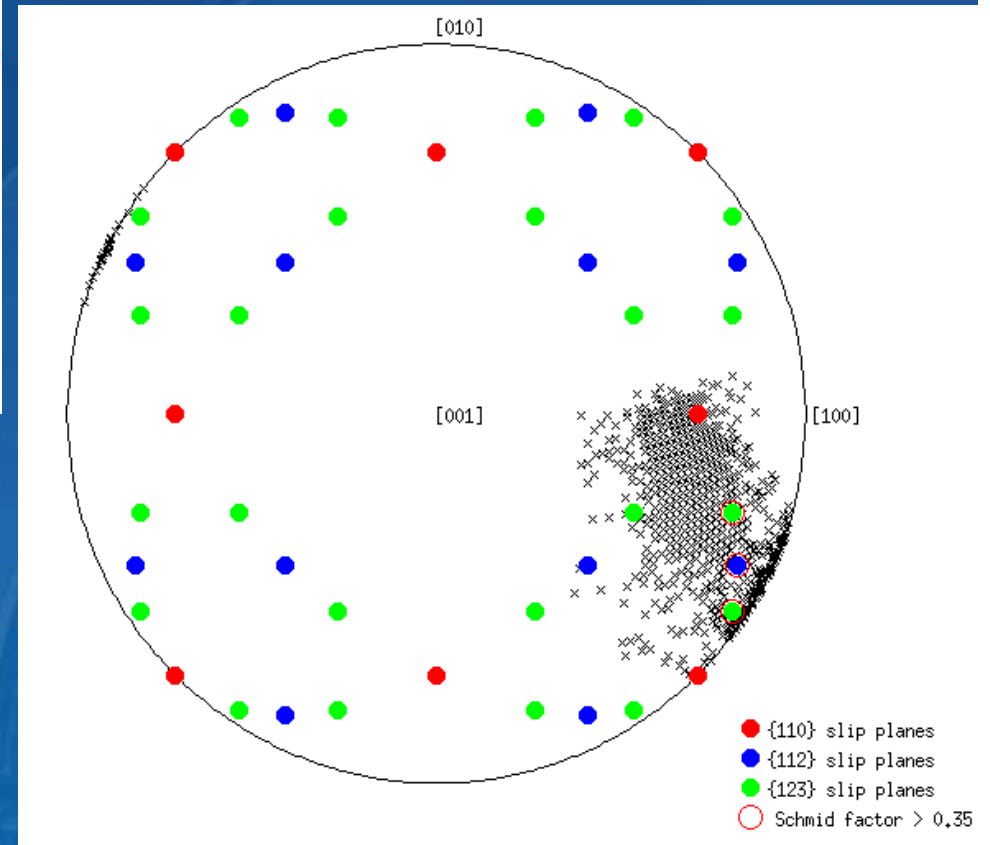
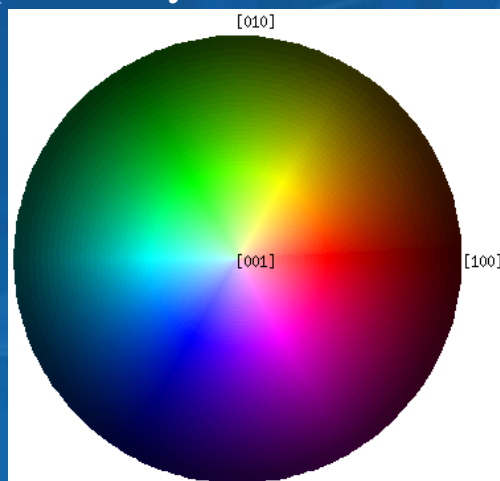
Orientation of crack in **crystal coordinate** system



Data Analysis – Orientation of crack in single grain



crack in single grain color coded with respect to crystal coordinate system



Conclusions

X-ray Diffraction Contrast Tomography can provide

- 3D grain shapes ~ 3 μm accuracy
- orientations ~ 0.1 degree
- elastic strain tensors $4 \cdot 10^{-4}$ (?)
- attenuation coefficient

in **plastically undeformed** polycrystals (single phase)
fulfilling some requirements on grain vs. sample size,
mosaicity and texture.

The technique can be readily combined with in-situ X-ray tomography observations and provides direct input for 3D Finite Element simulations

... **exciting times for 3D Materials Science**

Future directions / goals

- Further improve spatial resolution
 - detector point spread function
 - improved reconstruction algorithms :
 - fill gaps based on forward simulation
 - advanced algebraic reconstruction algorithms
- Characterization of **local orientation** and **strain state** in *deformed* materials
 - combine DCT & scanning microdiffraction approach
 - use of Monte Carlo simulation techniques

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James Marrow

TU Delft:

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