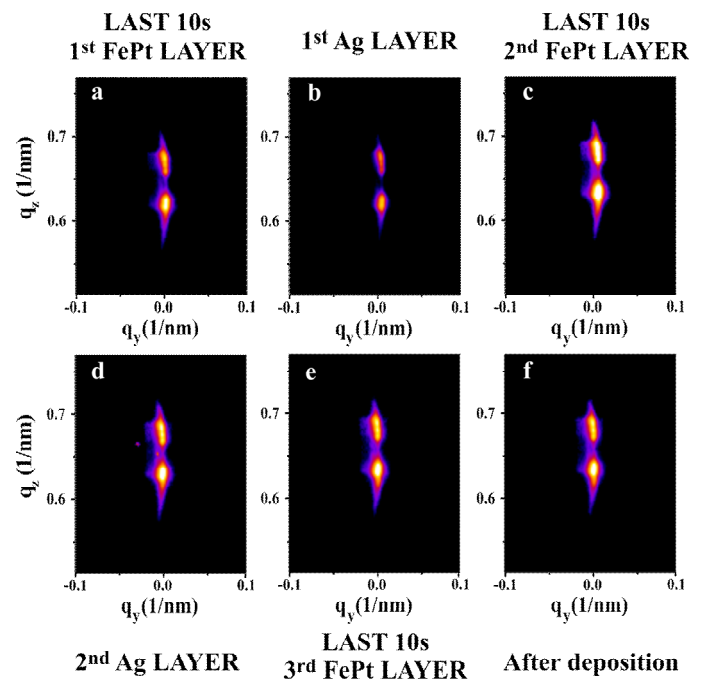
 ROBL-CRG	<b>Experiment title:</b> In-situ structural investigations of the L1 <sub>0</sub> ordering in FePt layers during magnetron sputtering	<b>Experiment number:</b> 20-02-665
	<b>Beamline:</b> BM 20	<b>Date of experiment:</b> from: 30.04.2008      to: 01.05.2008
<b>Shifts:</b> 9	<b>Local contact(s):</b> Dr. Nicole Jeutter	<i>Received at ROBL:</i>
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## REPORT

FePt is widely investigated because of its peculiar high magneto anisotropy density  $K_U = 3.7 - 6.2 \times 10^{19} \text{ eV/cm}^3$  [1] that allows to strongly reduce the volume of a single switching magnetic domain still satisfying the thermal stored data stability condition  $K_U V / k_B T = 50 - 70$  [1]. To ensure an acceptable noise reduction, FePt domains have to be magnetically decoupled, for example, by a diamagnetic matrix like Ag having a lower surface energy ( $\sim 7.11 \times 10^{14} \text{ eV/cm}^2$  [2]) than FePt ( $\sim 1.31 \times 10^{15} \text{ eV/cm}^2$  [3]) and therefore to not segregate into the FePt particles. RT deposited FePt layers exhibit a metastable soft - ferromagnetic A1 fcc phase. Subsequent annealing at temperatures above  $500^\circ\text{C}$  gives the tetragonal distortion to the strong ferromagnetic L1<sub>0</sub> phase [4].

In previous studies [5], we reported of low transition temperatures ( $T \sim 350^\circ\text{C}$ ) achieved using the ROBL dual magnetron deposition chamber [6]. In this study, we investigated FePt nanostructures evolution during deposition at  $400^\circ\text{C}$  as function of the Ag thickness (sample A: 7.3 nm thick Fe<sub>54</sub>Pt<sub>46</sub> layer separated by 0.2 nm Ag and sample B: 7.5 nm thick Fe<sub>57</sub>Pt<sub>43</sub> layer separated by 6.0 nm Ag) by in-situ grazing incident small-angle X-ray scattering (GISAXS) at an energy of  $E = 10 \text{ keV}$ . To detect the formation of the L1<sub>0</sub> FePt phase, a scintillation detector (BEDE) was positioned to analyse the (110) FePt superstructure peak at  $2\theta = 26^\circ$ .

Figure 1 shows the GISAXS patterns of sample A recorded during deposition at  $\alpha_i = 0.35^\circ$ , close to the Ag critical angle for total external reflection. During film growth,

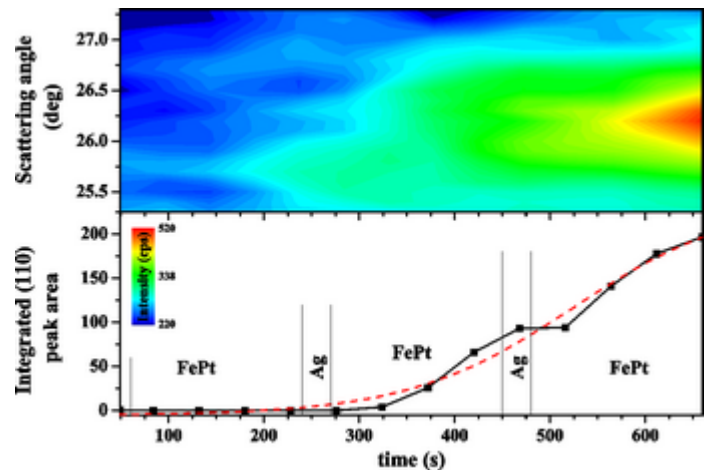


**Figure 1:** GISAXS evolution patterns collected for 10s during deposition at an incident angle  $\alpha_i = 0.35^\circ$  for sample A (intensity in logarithmic scale); figure (f) is collected just after deposition at  $400^\circ\text{C}$ .

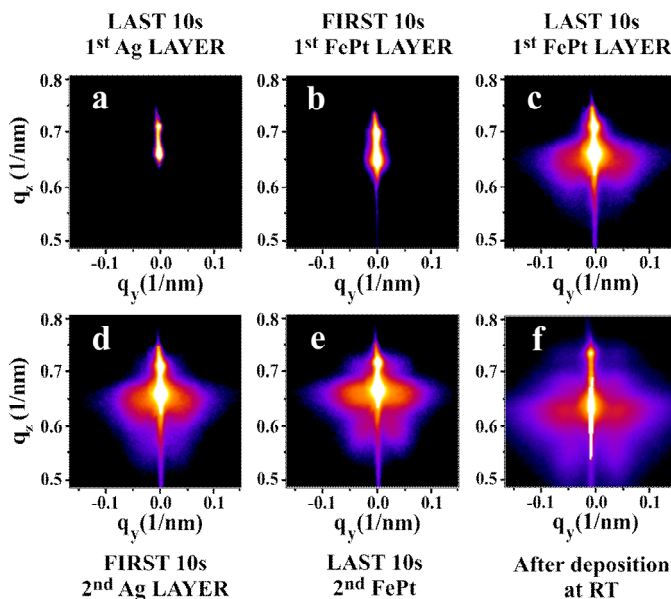
there is no significant change in the GISAXS patterns: specular and transmitted signals broadening points out only a roughness increasing. Ag amount is too low to separate FePt islands.

The FePt phase evolution, detected by the point detector, is reported in figure 2 as function of the deposition time. The L1<sub>0</sub> phase formation began around 300 s. Therefore, we obtained the L1<sub>0</sub> phase at 400°C with a minimum FePt thickness of about 12 nm.

In figure 3, the GISAXS evolution is shown for sample B: during growth figure 3a to figure 3e and after cooling down at RT, figure 3f. 6 nm of Ag directly on the a-SiO<sub>2</sub> at 400°C forms islands [7]. The subsequent FePt layer is, therefore, replicating the low frequency roughness created by Ag enhancing the GISAXS signal (figure 3c). Ag is forcing FePt agglomeration into clusters. From the GISAXS patterns collected at the end of deposition (fig. 3e, 3f) FePt clusters have side



**Figure 2:** GIXRD ( $\alpha_i = 0.5^\circ$ ) results on the (110) FePt peak during deposition of sample A. Lower graph: peak integrated area vs. deposition time (experimental data: black line with dots corresponding to the middle of each scan; simulation: red dashed line). Upper graph: (110) peak evolution vs. time.



**Figure 3:** GISAXS evolution patterns collected for 10s during deposition at an incident angle  $\alpha_i = 0.35^\circ$  for sample B (intensity in logarithmic scale): (a) is the profile at the end of the first Ag layer deposition, (b) is the beginning of the FePt deposition concluded in (c), (d) is the last Ag deposition and (e) the last FePt layer at 400°C; (f) is taken at RT directly after deposition.

facets tilted by almost 25° from sample surface normal and characteristic length size that shrinks at RT (fig. 3f) to 150 nm from 180 nm at the end of the first FePt deposition (fig. 3c).

The simultaneous detection of cluster growth and evolution by GISAXS and phase transition in XRD geometry was successfully applied to investigate, in-situ, the deposition of the FePt/Ag system on amorphous SiO<sub>2</sub> at a deposition temperature of 400°C. Using a certain amount Ag it is possible to trigger islands growth of the hard ferromagnetic L1<sub>0</sub>-FePt phase.

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